This article was downloaded by: On: *26 January 2011* Access details: *Access Details: Free Access* Publisher *Taylor & Francis* Informa Ltd Registered in England and Wales Registered Number: 1072954 Registered office: Mortimer House, 37-41 Mortimer Street, London W1T 3JH, UK



Liquid Crystals

Publication details, including instructions for authors and subscription information: http://www.informaworld.com/smpp/title~content=t713926090

Thermotropic liquid-crystalline polymers XXIII. Peculiarities of uniaxial orientation of comb-like liquid-crystalline polymers under mechanical stress

Ya. S. Freidzon^a; R. V. Talroze^a; N. I. Boiko^a; S. G. Kostromin^a; V. P. Shibaev^a; N. A. Platé^a ^a Moscow State University, Department of Chemistry, Moscow, U.S.S.R.

To cite this Article Freidzon, Ya. S., Talroze, R. V., Boiko, N. I., Kostromin, S. G., Shibaev, V. P. and Platé, N. A.(1988) 'Thermotropic liquid-crystalline polymers XXIII. Peculiarities of uniaxial orientation of comb-like liquid-crystalline polymers under mechanical stress', Liquid Crystals, 3: 1, 127 – 132 **To link to this Article: DOI:** 10.1080/02678298808086356

URL: http://dx.doi.org/10.1080/02678298808086356

PLEASE SCROLL DOWN FOR ARTICLE

Full terms and conditions of use: http://www.informaworld.com/terms-and-conditions-of-access.pdf

This article may be used for research, teaching and private study purposes. Any substantial or systematic reproduction, re-distribution, re-selling, loan or sub-licensing, systematic supply or distribution in any form to anyone is expressly forbidden.

The publisher does not give any warranty express or implied or make any representation that the contents will be complete or accurate or up to date. The accuracy of any instructions, formulae and drug doses should be independently verified with primary sources. The publisher shall not be liable for any loss, actions, claims, proceedings, demand or costs or damages whatsoever or howsoever caused arising directly or indirectly in connection with or arising out of the use of this material.

Thermotropic liquid-crystalline polymers

XXIII[†]. Peculiarities of uniaxial orientation of comb-like liquid-crystalline polymers under mechanical stress

by YA. S. FREIDZON, R. V. TALROZE, N. I. BOIKO, S. G. KOSTROMIN, V. P. SHIBAEV and N. A. PLATÉ

Moscow State University, Department of Chemistry, Moscow, 119899, U.S.S.R.

(Received 20 June 1986; accepted 6 June 1987)

The structure of oriented samples of liquid-crystalline side-chain polymers has been studied by X-ray diffraction. The arrangement of the different fragments of the macromolecules is shown to be dependent on the mesophase type of the initial melt. The orientation of the nematic melt leads to parallel alignment of mesogenic groups along the fibre axis. In the smectic melt, the smectic layers are arranged along the fibre axis with the mesogenic groups normal (S_A and S_B) or tilted (S_C and S_F) to it.

1. Introduction

The type of mesophase in liquid-crystalline side-chain polymers can be identified by the X-ray analysis of oriented fibres [1]. However the arrangement of the different fragments of the macromolecules in uniaxially oriented samples is not yet elucidated. Some new data have been discussed [2-8] but the limited number of polymers studied does not allow definite conclusions to be reached. It is necessary to note that in the papers [3-6] the structure of liquid-crystalline polymers oriented only by a magnetic field was studied. In this case the mesogenic groups are always arranged in the direction of the field applied. No special attention was given to the structure of liquid-crystalline polymers formed by drawing fibres from mesomorphic melts. Indeed, in this case the joint role of the main chain and the mesogenic groups defining the mesophase type should be essential.

Therefore in this paper we present the results of X-ray studies of a wide range of liquid-crystalline side-chain polymers

where R = H or CH_3 ; X = COO or CONH; n = 2 - 14

$$Y = -C00 - OC_{m}H_{2m+1}; - C00 - OC_{m}H_{2m+1}; - C00 - OC_{m}H_{2m+1}; - C00 - OC_{m}H_{2m+1}; - O - OC_$$

[†]Part XXII. FREIDZON, YA. S., BOIKO, N. I., SHIBAEV, V. P., and PLATÉ, N. A., 1986, *Eur.* Polym. J., **22**, 13. uniaxially oriented under mechanical stress. The synthesis of these polymers, their mesophase types, thermal properties and some structural data are described elsewhere [7–14]. Here we stress some general features of the orientation of liquid-crystalline polymers in the nematic and smectic phases.

2. Experimental

The oriented fibres were prepared by stretching from the liquid-crystal melts and were studied at room temperature. X-ray diffraction patterns were obtained with nickel-filtered CuK_{α} irradiation using a flat plate camera with a specimen-to-film distance of 60.5 mm.

2.1. Orientation of nematic polymers

Earlier we showed [7] the possibility of forming two types of nematic structure in polymers having side mesogenic groups. The X-ray diffraction patterns of the first type, having only azimuthal order of the mesogenic groups, are characterized by one diffuse halo at wide angles. The same X-ray diffraction patterns are observed for low molar mass nematics. The second type of nematic order is characterized both by azimuthal and hexagonal order in the plane normal to the nematic director. The corresponding X-ray diffraction patterns have one sharp diffractional maximum at wide angles without any small-angle reflections. We call this nematic mesophase the N_B phase unlike the usual N_A phase [7].

The uniaxial orientation of all nematic polymers under study is accompanied by a splitting of the wide-angle maximum into two crescents on the equator. The X-ray diffraction patterns of polymers (1) and (2) forming nematic phases A and B at room temperature are shown in figures 1 (a) and (b).

$$\begin{array}{c} -CH_2 - CH_2 \\ c \\ 00 - (CH_2)_5 - C00 \\ N_B & 60^{\circ}C \\ N_A & 120^{\circ}C \\ 1 \end{array}$$
(2)



Figure 1. X-ray diffraction patterns of uniaxially oriented polymers (a) (1) and (b) (2) in (a) N_A and (b) N_B phases.



Figure 2. X-ray diffraction patterns of uniaxially oriented polymers (a) (3) and (b) (4) in (a) S_A and (b) S_C phases.

2.2. Orientation of smectic polymers

Among smectic polymers having side mesogenic groups smectic phases with both two-dimensional order (S_A and S_C) and three-dimensional order (S_B and S_F) have been observed. The polymers in smectic A and C phases are easily oriented by stretching the anisotropic melt. The X-ray diffraction patterns of polymers (3) and (4) oriented in their S_A and S_C phases are shown in figures 2 (*a*) and (*b*). The polymers existing in

CН

ÇH₃

$$-CH_{2} - CH_{2} - CH_{2} - CN = CN$$

$$-CH_{2} - CH_{2} - CN = CN$$

$$S_{A}^{121^{\circ}C} I = (3)$$

$$\begin{array}{c} CH_2 - CH_2 \\ L \\ COO - (CH_2)_{11} - 0 & \bigoplus CN \\ S_C \ 30^{\circ}C \ S_A \ 145^{\circ}C \ I \end{array}$$
(4)

some three-dimensional ordered mesophase cannot be transformed into an oriented state. Their orientation is possible only in two-dimensional ordered smectic phases which are formed at higher temperatures or in the nematic phase. In such cases the transition into the three-dimensional ordered smectic phase proceeds by cooling. Typical X-ray diffraction patterns of oriented samples obtained in this way for polymers (5)-(7) are shown in figure 3.

$$\begin{array}{c} - CH_2 - \dot{C} - \\ COO - (CH_2)_{11} - 0 \longrightarrow CH_2 N \bigoplus C_4 H_9 \\ S_B 86^{\circ}C S_A 140^{\circ}C 1 \end{array}$$
(5)

$$CH_2 - CH_2 - CH_2 - COO - OC_4H_9$$
 (6)
 $S_F 45°C S_C 145°C 1$

$$CH_2 - CH_2$$

 $COO - (CH_2)_5 - COO - OO - OO - OC_4H_9$ (7)
 $S_F 70°C N_A 135°C I$



Figure 3. X-ray diffraction patterns of uniaxially oriented polymers (a) (5), (b) (6) and (c) (7) in smectic phases (a) S_B and $(b, c) S_F$. The oriented fibres of polymers (6) and (7) were prepared by stretching from S_C and N_A phases, respectively.

We can see from figures 2 and 3 that the X-ray diffraction pattern of samples obtained by orientation in S_A and S_C mesophases exhibit small-angle reflections which are always split into two equatorial maxima, while the wide-angle reflections are split either into two crescents on the meridian (for the phases having an orthogonal arrangement of the mesogenic groups, S_A or S_B) or into four crescents (for the tilted arrangement of the mesogenic groups, S_C or S_F). The X-ray diffraction patterns of polymer (7) which forms the smectic phase upon cooling of the sample oriented in the nematic phase consists of two wide-angle crescents and four point-like reflections at small angles. The additional splitting of one of the small-angle reflections into four crescents for polymer (4) is related to the coexistence of two types of layer packing for the mesogenic groups [13, 14].

3. Discussion

The wide-angle maximum in the X-ray diffraction patterns of polymers having side mesogenic groups is known [1-7, 9] to be related to the lateral packing of the mesogenic groups. The small-angle reflections correspond to the thickness of the smectic layers.

The orientation of polymers from the nematic mesophase is accompanied by the arrangement of the side mesogenic groups along the fibre axis. Such behaviour of nematic polymers in a mechanical field is a result of features of the deformation of liquid-crystalline polymers characterized by the existence of a region of equally oriented mesogenic groups (domains). The action of various fields (electric, magnetic, mechanical) on low molar mass liquid crystals is known to result in the orientation of the domains. For the polymers under investigation the domains are formed by the mesogenic side groups. Under orientation the long axes of these groups are disposed along the fibre axis. The backbones probably have a random conformation. A schematic diagram of the structure of an oriented nematic polymer is shown in figure 4(a).

The orientation of smectic polymers promotes the slipping of smectic layers and results in their disposition along the direction of the mechanical field. The orientation of the mesogenic groups depends on the type of the smectic mesophase: for S_A and S_B phases the mesogenic groups are disposed normal to the smectic layers and to the fibre axis (cf. figure 4(*b*)), and in the S_C phase they are tilted (cf. figure 4(*c*)).



Figure 4. Schemes of macromolecular packing in oriented polymers in (a) N_A and N_B , (b) S_A and S_B , (c) S_C and S_F (oriented in S_C) and (d) S_F (oriented in N_A) phases.

The arrangement of smectic layers and mesogenic groups in the S_F phase depends on the structure of initial melt. Thus, orientation of polymer (6) is possible in the S_C phase and is accompanied by disposition of smectic layers along the fibre axis, while the angle between the mesogenic groups and the fibre axis is equal to 34°. The transition to the S_F phase is accompanied by ordering of the mesogenic groups in layers, while their orientation is kept constant. The orientation of polymer (7) proceeds in the nematic phase, and as a result the mesogenic groups are disposed along the fibre axis. The transition into the S_F phase is accompanied by displacement of mesogenic groups along their long axes keeping the orientation and by the formation of layers tilted to the fibre axis (cf. figure 4(d)).

As the backbones of macromolecules are disposed mainly in the planes of smectic layers their direction should presumably coincide and be disposed along the fibre axis.

In analysis of the splitting of the wide-angle maximum into the equatorial crescents in the X-ray diffraction patterns of oriented nematic comb-like polymers, the authors of [2] have doubted the possibility of the alignment of the mesogenic groups along the fibre axis because of the impossibility of normal disposition of the backbone. An alternative scheme was proposed which included the parallel disposition of backbones and normal disposition of the mesogenic groups. In such a case the macromolecules look like a planar ribbon and the wide-angle diffraction maximum is related to the reflection from these parallel ribbons. This assumption contradicts the well-known experimental data [1–7, 9] on the structure of polymers having side mesogenic groups, according to which the wide-angle reflection corresponds to the distance between the neighbouring mesogenic groups.

Thus the results of this work show the essential dependence of the disposition of the backbones and the mesogenic groups on the type of the polymer mesophase during uniaxial orientation under mechanical stress of comb-like liquid-crystalline polymers.

References

- [1] SHIBAEV, V. P., and PLATÉ, N. A., 1984, Adv. Polym. Sci., 60/61, 173.
- [2] ZENTEL, R., and STROBL, G. R., 1984, Makromolek. Chem., 185, 2669.
- [3] ZUGENMAIER, P., and MÜGGE, J., 1984, Makromolek. Chem. rap. Commun., 5, 11.
- [4] DAVIDSON, P., KELLER, P., and LEVELUT, A. M., 1985, J. Phys., Paris, 46, 939.
- [5] DECOBERT, G., SOYER, F., DUBOIS, J. C., and DAVIDSON, P., 1985, Polym. Bull., 14, 549.
- [6] DIELE, S., HISGEN, B., RECK, B., and RINGSDORF, H., 1986, Makromolek. Chem. rap. Commun., 7, 267.
- [7] FREIDZON, YA. S., BOIKO, N. I., SHIBAEV, V. P., and PLATÉ, N. A., 1985, Dokl. Akad. Nauk SSSR, 282, 934.
- [8] PLATÉ, N. A., FREIDZON, YA. S., and SHIBAEV, V. P., 1985, Pure appl. Chem., 57, 1715.
- [9] SHIBAEV, V. P., FREIDZON, YA. S., and PLATÉ, N. A., 1976, Dokl. Akad. Nauk SSSR, 227, 1412.
- [10] FREIDZON, YA. S., SHIBAEV, V. P., KUSTOVA, N. N., and PLATÉ, N. A., 1980, Vysokomolek. Soedin. A, 22, 1083.
- [11] FREIDZON, YA. S., KHARITONOV, A. V., SHIBAEV, V. P., and PLATÉ, N. A., 1985, Eur. Polym. J., 21, 211.
- [12] TALROZE, R. V., SINITZYN, V. V., SHIBAEV, V. P., and PLATÉ, N. A., 1982, Molec. Crystals liq. Crystals, 80, 211.
- [13] KOSTROMIN, S. G., SINITZYN, V. V., TALROZE, R. V., SHIBAEV, V. P., and PLATÉ, N. A., 1982, Makromolek. Chem. rap. Commun, 3, 809.
- [14] KOSTROMIN, S. G., SINITZYN, V. V., TALROZE, R. V., and SHIBAEV, V. P., 1984, Vysokomolek. Soedin. A26, 335.